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New Organic-Inorganic Nanocomposite Materials for Energy

Storage Applications

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Abstract

A method is described by which the disulfide dimer of 2,5-dimercapto-1,3,4thiadiazole (diDMcT, see Scheme 1) can be oxidatively intercalated into the layered structure of a V₂O₅ xerogel. This intercalation reaction produces a new organic-inorganic composite material with a layer spacing of 13.5 Å, in contrast to the 11.55 Å spacing for the parent V₂O₅ xerogel. During this oxidative intercalation, the diDMcT is polymerized to produce a polymer with thiadiazole rings linked by disulfides in the polymer main chain (PDTT, see Scheme 1). The composite material is characterized by uv-visible spectroelectrochemistry, x-ray diffraction, FTIR and electrochemistry. The electrochemical experiments comprised charging (oxidation) and discharging (reduction) of the material, with the bulk of the redox reaction occurring over a broad potential range of 0.5 V to -0.6 V versus SCE. The cyclic voltammogram of the composite material shows features that can be attributed to the DMcT-PDTT redox response. However, during or after reduction of the composite, the monomeric DMcT dithiolate appears to be expelled from the V2O5 interlayer region, leading to an evolution of the electrochemical response back to that of the original V₂O₅ material. Evidence is presented suggesting that the V₂O₅ host material facilitates the redox reactions of the thiol-disulfide redox couple while it is within the interlayer region.

Introduction

Materials that are capable of rapid and efficient charge storage and delivery (i.e. rechargeable redox couples) are important in a host of applications. In many cases, these materials are optimized for a particular aspect of their behavior. For example, the inorganic oxides that have been widely used as Li secondary battery cathode materials¹⁻⁴ have good

stability toward long term cycling and fairly positive redox potentials. On the other hand, they suffer from energy densities that are well below those that can be attained with available anodes, and they have relatively low power densities. ¹⁻⁴ These detrimental characteristics have driven a high level of recent activity in the development of new cathode materials. ⁴⁻¹² One of the materials that has emerged as having an attractive combination of properties is V₂O₅. Both high energy density and good reversibility have been demonstrated for this material, ⁹⁻¹² especially in its aerogel form. ^{9,11,12}

In battery applications, discharge (reduction) of these materials is accompanied by Li⁺ insertion, which is often rate-limiting. Thus, a common strategy to achieve higher discharge rates is to use high surface area materials, so that the distance over which Li⁺ must diffuse through the host material is minimized. Another approach that does not appear to have been explored yet is to manipulate the interlayer spacing in these layered materials by using various intercalants so as to enhance the rate of Li⁺ migration. It is well known that intercalation leads to changes in the interlayer spacing for layered materials. Examples of such intercalants include thiophene, pyrrole and aniline, all of which are known to oxidatively polymerize when intercalated into highly oxidizing materials such as V₂O₅. Much of the previous work has focussed on electronic and optical properties of the resulting "nanocomposites." However, little has been done on the influence of intercalation on the redox reactions of either the host material or the intercalant. One goal of our research effort is to examine these effects, especially with regard to the influence of intercalants on Li⁺ diffusion rates and charge capacity in these nanocomposites.

In the present case, the intercalant that has been investigated comprises 2,5-dimercapto-1,3,4-thiadiazole (DMcT) in its various monomeric and oligomeric redox forms. Scheme 1 shows the various redox states of DMcT. As has been previously discussed, ^{37, 39} the first oxidation product of DMcT is diDMcT (reaction 1). This disulfide dimer can be oxidatively polymerized to produce PDTT, which is an oligomeric material of undefined degree of polymerization and very low solubility in most common solvents. This material was originally investigated for charge storage purposes by De Jonghe, Visco and coworkers. ^{28, 29} They found it to be unsuitable for use at ambient temperatures because of

sluggishness of the thiol-disulfide redox couple. However, more recently, Oyama and coworkers found that its redox reactions could be "facilitated" by making molecularly mixed composites of DMcT and poly(aniline). 30-38 In order to better understand this phenomenon, we have recently conducted several detailed mechanistic and spectroscopic studies of the redox behavior of DMcT and several of its derivatives, including studies of its acid-base reactions. 37, 39 These studies have suggested that the facilitation of DMcT redox cycling is likely a result of proton transfer between poly(aniline) and the DMcT oligomer (PPDT, see Scheme 1). This is especially true during reduction, where it is believed that the availability of protons (from poly(aniline)) drives reductive cleavage of the S-S bond at less negative potentials than would otherwise be observed. In this regard, one of the motivations of the present study was to examine whether or not such redox facilitation could be achieved by using transition metal centers in host materials such as V₂O₅. As will be seen below, in spite of long term instability of the PDTT-V₂O₅ composite with respect to loss of DMcT, there is evidence for this facilitation by the host material, suggesting that similar strategies with other materials might be worthy of additional investigation.

Experimental

Materials

All chemicals were of reagent grade or better and were used as received. Acetonitrile (ACN) was freshly distilled from P₂O₅ prior to use. The vanadium oxide xerogel solution was produced as previously described by ion exchange of NaVO₃ to produce HVO₃, followed by aging for at least two weeks. 40-42 This procedure produces a viscous, deep wine red solution. DiDMcT was prepared by slow air oxidation of DMcT in ethanol/water (50/50 v/v), producing lemon yellow crystalline needles (yield = 89%). Crystallization started after about 20 hours, and the reaction mixture was left for three days prior to harvesting the crystals. Elemental analysis gave: (formula: C₄H₂N₄S₆) calculated: H 0.68, C 16.11, N 18.8, S 64.4, observed: H 0.68, C16.19, N 18.78, S 64.35. The structure was confirmed by x-ray diffraction (see supplementary material), and shows that diDMcT exists in the thioamide tautomer (as opposed to the dithiol tautomer shown in Scheme 1) in the solid state. 37, 39

Equipment and procedures

Uv-vis spectroscopic measurements were made using a HP 8452 diode array spectrometer with optically transparent indium-doped tin oxide (ITO) electrodes (Delta Technologies). Electrochemical measurements were made using a PAR 273. X-ray diffraction was done on a Scintag XDS2000 using the following conditions: Cu Kα, 40 KV, 30 mA, continuous scan, 1° 2θ/min, 0.01° 2θ/increment (0.60 seconds/point). Spin coating of the neat gel solution was done using a locally modified unit at a rotation rate of 5000 RPM. FTIR transmission measurements were made on a Bomem MB100 after pressing the samples in KBr pellets.

Films cast onto various electrode substrates had relatively poor adhesion unless the film thickness was kept below ca. 1500 Å. For this reason and to achieve better film uniformity and homogeneity, a spin coating procedure was developed. Spin coated films were prepared on either ITO electrodes or on vapor deposited thin film Au electrodes (2000 Å in thickness, electrode area of 0.34 cm²) and dried at 120° C for 20 hours at 0.75 atm (ambient pressure), producing a dark red film. Films prepared in this way were stable for over 50 cyclic voltammetric cycles before adhesive failure led to film delamination. Film thicknesses were typically 1500 Å and were measured on a Rudolph AutoEl II ellipsometer at a wavelength of 632.8 nm using the following values for refractive index: 1.76 (real part) and 0.0 (imaginary part), which are values for crystalline V₂O₅. Because the refractive index values for V₂O₅ xerogels have not been reported, we consider the film thicknesses obtained in this way as fairly rough estimates. Note that the lack of absorption of the V₂O₅ films at 632.8 nm (see below) supports the use of 0.0 for the imaginary part of the refractive index.

DiDMcT was oxidatively intercalated into V_2O_5 by exposing V_2O_5 films (prepared as described above) to 5 mM ACN solutions of diDMcT for 48 hours at reflux. Oxidative intercalation of diDMcT and simultaneous reduction of vanadium sites in the host material were characterized by a color change of the film from deep red to dark green, as expected for reduction of V(V) to V(IV). Initial experiments using DMcT as the intercalant

produced composite films that had very poor redox and adhesive properties. We suspect this is due to the strongly acidic nature of DMcT ($pK_{a1} = -1.36$),³⁷ which may adversely affect the V_2O_5 host material. Thus, all experiments were done with composite films produced by oxidative intercalation of diDMcT into V_2O_5 . As noted above, for short term experiments (i.e. less than 50 scans) the adhesion of these films to ITO or Au electrodes was not adversely affected by the dimensional changes accompanying the intercalation (see below), so long as the film thickness was ca. 1500 Å or less.

The supporting electrolyte for all experiments was 0.2 M LiClO₄ in ACN. Cyclic voltammetric experiments with diDMcT and DMcT in 0.2 M LiClO₄/ACN solution were made using a glassy carbon working electrode with an area of 0.0707 cm², a Pt counter electrode and a Ag/AgCl reference electrode, with respect to which all potentials are reported. Solutions were thoroughly purged with Ar prior to all experiments.

Results and Discussion

Figure 1a shows the results of a spectroelectrochemical experiment in which a thin film of V_2O_5 on an ITO substrate was sequentially exposed to potentials between 1.3 V and -0.2 V. Reduction of the V(V) sites in the lattice to V(IV) leads to loss of the strong band at 385 nm and simultaneous appearance of a broad, weaker long wavelength band with λ_{max} near 750 nm. These data are entirely consistent with previous observations of the spectral changes accompanying redox reactions of these materials. Figure 1b (curve b) shows the uv-vis spectrum that is observed when a V_2O_5 film on an ITO substrate is exposed to a diDMcT solution in ACN. Reaction results in loss of the strong 385 nm band and the appearance of weak absorbance in the long wavelength region above 500 nm. These spectral changes are consistent with the reduction of V(V) to V(IV) during exposure to diDMcT. Based on these results, it is reasonable to assume that diDMcT must be oxidized during this process. Unfortunately, the uv cutoff of the ITO substrate does not allow direct observation of the spectral changes that should accompany this oxidation. 37,39

FTIR characterization of V_2O_5 films before and after exposure to diDMcT is also helpful in diagnosing the reaction between them. Figures 2a and 2b show the infrared

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spectra of diDMcT and PDTT, respectively. As can been seen, both compounds show strong bands near 1050 cm⁻¹ and 1380 cm⁻¹. However, they can be distinguished from each other by virtue of the presence of a strong doublet in the 1275 cm⁻¹ region for diDMcT and the lack of any significant bands in this region for PDTT. Assignment of the infrared and Raman spectra for these compounds is the subject of an impending contribution,⁴⁴ and is not necessary for the purposes of the present investigation. Figure 2c shows the spectrum of a V₂O sample. The V-O-V and V=O bands at ca. 800 cm⁻¹ and 1050 cm⁻¹, respectively, are characteristic of this material, 43 as is the weak band at 1640 cm⁻¹ showing the presence of some residual water in the sample. 43 Figure 2d shows the spectrum that is obtained for a V_2O_5 sample that was exposed to 5 mM diDMcT in ACN at reflux for 48 hours. This spectrum very clearly shows that the sample contains a large amount of PDTT, based on the presence of strong bands near 1050 cm⁻¹ and 1380 cm⁻¹, and no diDMcT, based on the complete absence of any bands in the 1275 cm⁻¹ region. The ratio of the PDTT band intensities to those for the V₂O₅ component of the sample suggests that the sample contains a relatively large amount of PDTT relative to V₂O₅. It is possible to use time and temperature to control the amount of PDTT produced by this reaction. For example, exposure for 48 hours at ambient temperature, rather than at reflux, produces a sample that has relatively weaker PDTT bands and relatively stronger V₂O₅ bands (data not shown). Based on previous reports that V_2O_5 is swollen by ACN at reflux temperatures, ²³ these results suggest that the amount of PDTT produced in this reaction is determined by how the reaction conditions influence access of diDMcT to the interior of the V₂O₅ layers.

X-ray diffraction of spin coated films of V₂O₅ and PDTT-V₂O₅ shows prominent reflections at 2θ values that give lattice spacings of 11.55 Å and 13.5 Å, respectively (data not shown). This value for the V₂O₅ xerogel is consistent with previous observations for this material, and suggests that the films produced by the spin coating-heat treatment procedure described in the Experimental section results in V₂O₅•1.5 H₂O.⁴⁰⁻⁴² The value for the PDTT-V₂O₅ samples is consistent with the premise that the PDTT is intercalated into the interlayer region of the V₂O₅ host lattice, based on previous observations of changes in layer spacing on intercalation of other planar organic compounds into layered materials.¹⁸⁻²⁷ Interestingly, the XRD data for the PDTT-V₂O₅ sample do not show a peak corresponding to

the 11.55 Å spacing, suggesting that the intercalation reaction is complete and that none of the V₂O₅ parent material remains after reaction. Thus, taken together, the data suggest that the reaction between diDMcT and V₂O₅ produces a material with a structure such as that schematically depicted in Scheme 2, in which the PDTT oligomers are intercalated into the interlayer region of the V₂O₅ host lattice and the planes of the thiadiazole rings are nearly parallel with the V₂O₅ layers. A non-parallel orientation of the thiadiazole rings would not be consistent with the layer spacing of 13.5 Å. In this regard, it is interesting that the crystal structure of diDMcT (see supplementary material) shows the C-S-S-C dihedral angle to be 92°, as expected. Additionally, the angle between the planes of the two thiadiazole rings in each dimer is ca. 90°. This geometry leads to unit cell dimensions of 5.72 Å x 5.92 Å x 15.89 Å, with the two shorter dimensions being roughly determined by the distance across the two thiadiazole rings. As mentioned above, such a geometry would *not* be consistent with the interlayer spacing of 13.5 Å for the PDTT-V₂O₅ composite, suggesting that the angle between the two thiadiazole rings in this material must be considerably less than 90°.

Figure 3a shows a cyclic voltammetric experiment for a V₂O₅ thin film on a vapor deposited Au electrode. The broad cathodic and anodic waves in the range 0.5 V to -0.6 V result from the reduction and oxidation, respectively, of V sites in the lattice. These data are in excellent agreement with previous reports on V₂O₅ xerogels. ⁹⁻¹¹ A detailed analysis of the voltammetry of V₂O₅ xerogels is not attempted here, being unnecessary for the purposes of the present discussion. Figure 3b (solid curve) shows the first reductive scan for a PDTT-V₂O₅ thin film. The salient feature is a much larger cathodic current at ca. -0.1 V compared to the response of the V₂O₅ xerogel parent material. Some of this cathodic charge is recovered in the same potential range during the positive-going, return scan. We attribute this excess charge to the reduction and reoxidation of some of the PDTT material that is present within the interlayer region. This charge is considerably diminished in the second scan (dashed curve) and continues diminishing during subsequent scans. With continuous scanning, the voltammetry of the system slowly (ca. 1 hour) evolves back to that of the V₂O₅ xerogel parent material. This suggests that some amount of the DMcT (which most likely will be present as the dithiolate³⁷⁻³⁹) that is produced on reduction of the PDTT oligomer

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diffuses out of the V_2O_5 host lattice during each scan. After many scans, the voltammetry indicates that most of the DMcT/PDTT intercalated material has been lost from the V_2O_5 thin film.

It is interesting to compare the behavior of the DMcT-PDTT redox couple within the V₂O₅ lattice to that observed for diDMcT and DMcT in solution. Figure 4 shows the cyclic voltammetry for a diDMcT sample in ACN solution. This compound shows a cathodic peak at -0.2 V which corresponds to reductive cleavage of the S-S bond in diDMcT to give the monothiolate form of DMcT. 37, 39 The anodic peak at 0.15 V corresponds to reoxidation of the DMcT monothiolate to give diDMcT. 37,39 The situation is more complex for DMcT, due in large measure to polymerization-depolymerization processes and proton transfer that occur during the redox transitions. 37, 39, 45 Figure 5a shows the voltammetry for a solution of DMcT at a scan rate of 200 mV s⁻¹. At this scan rate extensive precipitation and corresponding complications in the voltammetry are somewhat avoided; lower scan rates lead to much more complex voltammetry. The salient feature of Figure 5a are the peaks for DMcT oxidation at potentials in the range 0.8 V to 1.2 V and for the corresponding reduction at -0.4 V. This oxidation potential is that observed for the neutral, fully protonated form of DMcT. 37, 39, 45 Note that it is nearly a volt more positive than that for the oxidation of the DMcT monothiolate form, showing the very strong influence of proton availability (or lack of it) on redox potentials in this non-aqueous solvent system. The subsequent reduction is that observed for direct, "unassisted" (meaning unassisted by protons, see below) reductive cleavage of the S-S bond in PDTT. Figure 5b shows the cyclic voltammetric response that is observed when a strong acid (trifluoromethane sulfonic acid, TFMSA) is added to the solution. The important feature to note is the appearance of a new, less negative reduction wave at ca. 0.0 V. As has been established in other studies, this wave corresponds to the proton assisted (or facilitated) reductive cleavage of the S-S bond. 37, 39, 45 Comparison of the potential for PDTT reduction in the presence of protons with that for PDTT reduction in the V₂O₅ lattice reveals a good correspondence between the two. It is especially significant that these reductions occur ca. 0.3 V less negative than that for the unassisted reductive cleavage of the S-S bond in PDTT. This suggests that the metal

centers in the V_2O_5 lattice may act in a manner similar to that of protons in terms of facilitating the S-S reductive cleavage.

Conclusions

This study has revealed that diDMcT can be oxidatively intercalated into the interlayer region of a V₂O₅ xerogel host lattice, as confirmed by uv-vis and FTIR spectroscopic studies, x-ray diffraction and electrochemical experiments. This redox reaction produces a material whose composition is consistent with an organic-inorganic composite that can formulated as $(PDTT)_x$ - $(V_2O_5)_y$, where x and y can be controlled by the reaction conditions. While intercalation is relatively facile and can be driven to high levels, the electrochemical behavior of the composite material during redox cycling suggests that DMcT is lost from the V₂O₅ interlayer region while in its reduced form, most likely due to diffusion out of the lattice. This may not be unexpected, given the electrostatic repulsion expected between the negatively charged, reduced V_2O_5 lattice⁴⁰⁻⁴² and the monothiolate or dithiolate forms of DMcT (which are the most likely forms to exist under these conditions). 37,39 However, in spite of this loss of intercalant, the suggestion that the V centers may be facilitating the S-S reductive cleavage indicates that further studies of composite materials of S-containing guests in transition metal hosts may be worthwhile. This is especially true since the kinetics, rather than the thermodynamics, of the thioldisulfide redox transformations appear to be the stumbling block for their use in redox cycling applications. 28, 29

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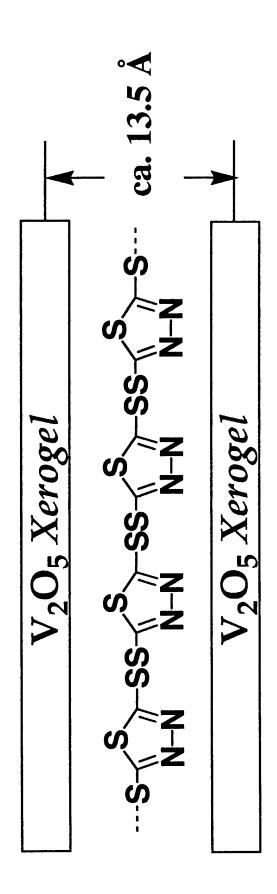
Figure captions

- a) In situ spectroelectrochemical experiment on a V₂O₅ thin film on ITO. The indicated potentials were applied until the spectral changes were complete, typically ca. 5 min. b) Curve a: ex situ spectrum of a V₂O₅ thin film on ITO. Curve b: ex situ spectrum of the same film as in Curve a, after exposure to diDMcT as described in the Experimental section.
- 2. FTIR spectra of pressed KBr pellets of: a) DMcT, b) PDDT, c) V₂O₅, and d) PDTT-V₂O₅ (see text for details).
- 3. Cyclic voltammograms of: a) a V₂O₅ thin film on Au, and b) a PDTT-V₂O₅ thin film on Au (first scan solid, second scan dashed), scan rates = 5 mV s⁻¹.
- 4. Cyclic voltammogram of 5 mM diDMcT, scan rate = 50 mV s⁻¹. Steady state response, achieved after multiple scans.
- 5. Cyclic voltammograms of: a) 10 mM DMcT (repetitive cycles with increasingly positive potential limit), and b) 10 mM DMcT + 2 mM TFMSA (repetitive cycles with increasingly positive potential limit); scan rates = 200 mV s⁻¹.

$$HS \underset{N-N}{\swarrow} SS \underset{N-N}{\swarrow} SH \qquad (1)$$

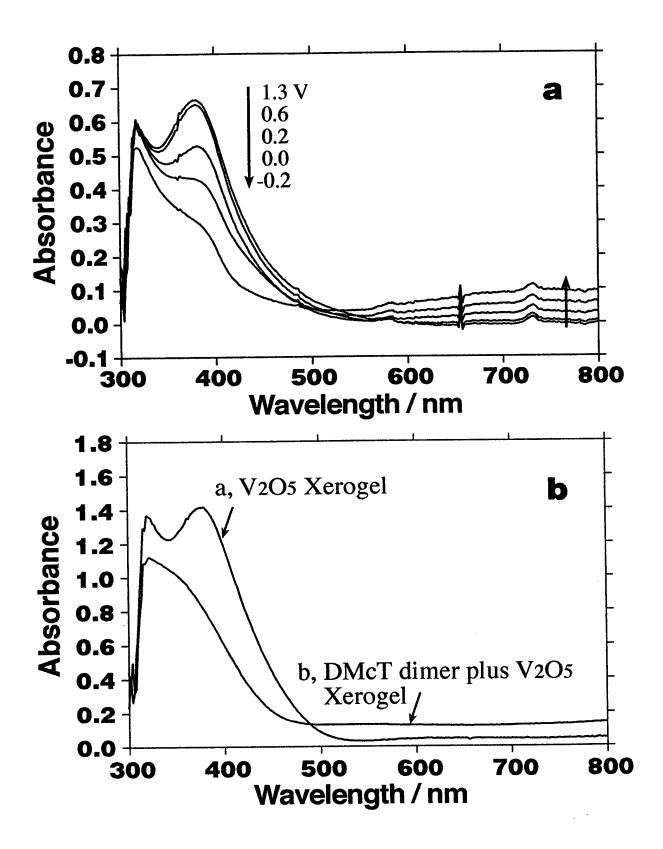
$$DMcT Dimer$$

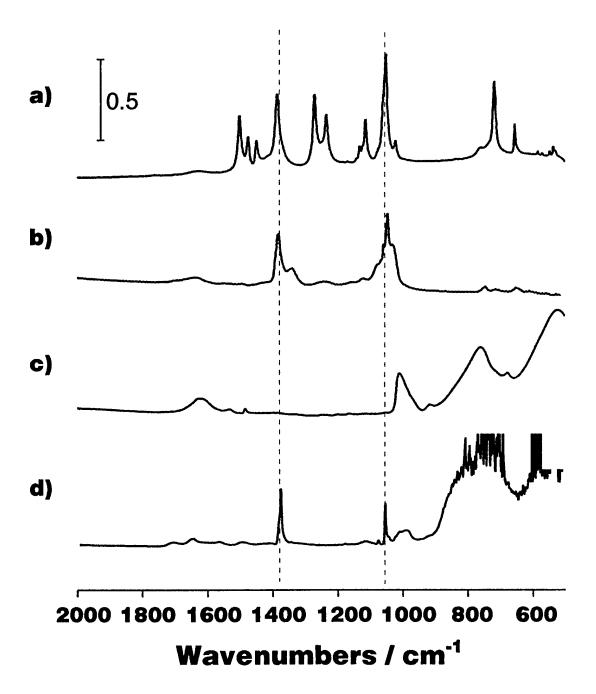
$$PDTT-V_2O_5$$
 hybrid

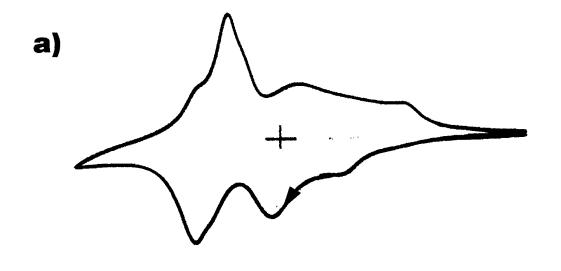


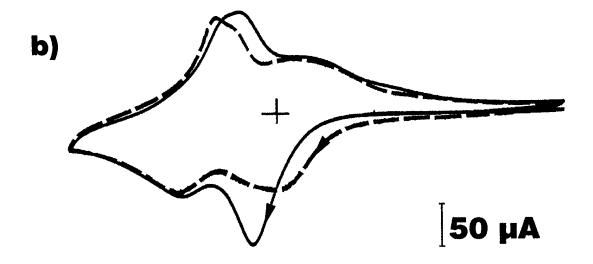
Scheme 2

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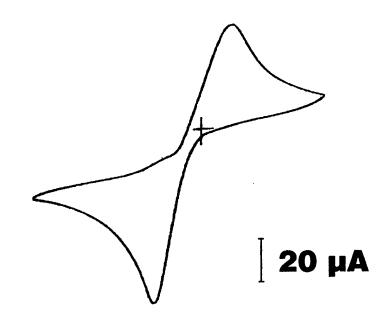




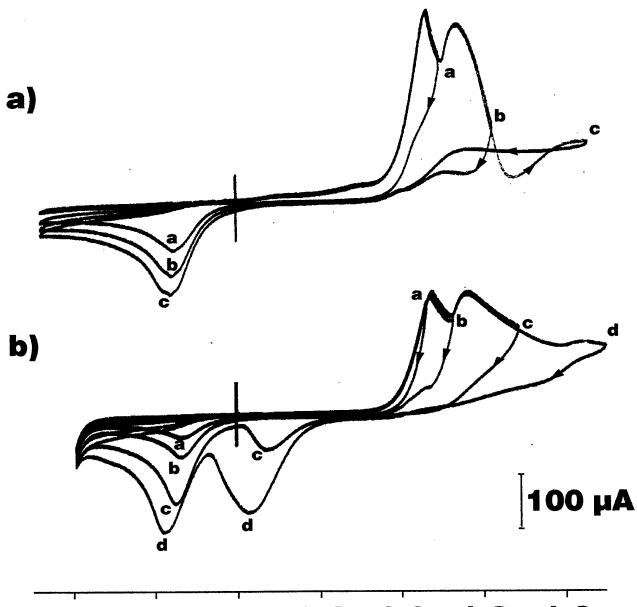




-0.8 -0.4 0.0 0.4 0.8 1.2 E / V vs. Ag/AgCl



-0.8 -0.4 0.0 0.4 0.8 E / V vs. Ag/AgCl



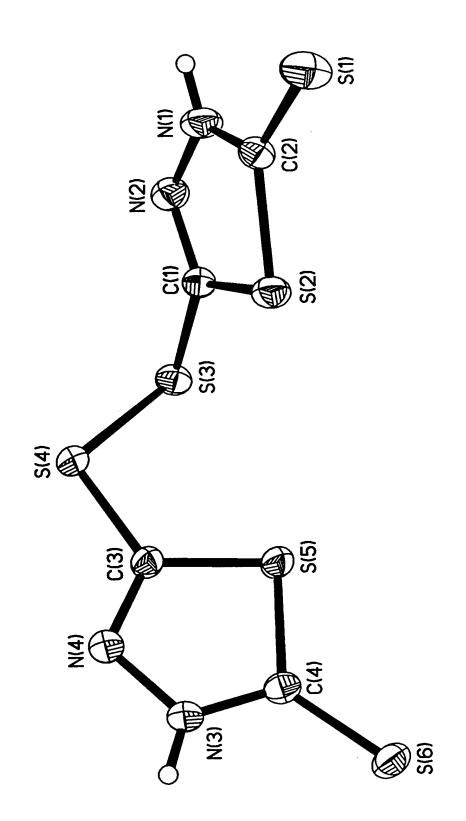
-0.8 -0.4 0.0 0.4 0.8 1.2 1.6 E / V vs. Ag/AgCl

Supporting Material

Single crystal X-ray data were collected at 25 °C using a yellow crystalline plate of dimensions $0.27 \times 0.22 \times 0.06$ mm on a Siemens P4 diffractometer equipped with a molybdenum tube ($\lambda = 0.71073$ Å) and a graphite monochromator. A cell of dimensions a = 5.7200(10) Å, b = 5.9170(10) Å, c = 15.889(2) Å, $\alpha = 94.410(10)^{\circ}$, $\beta = 98.440(10)^{\circ}$, $\gamma = 104.460(10)^{\circ}$, and V = 511.46(14) ų was determined using the measured positions of 41 reflections in the 20 range 11 to 35°. A total of 2370 reflections were gathered, the octants collected being +h, $\pm k$, $\pm l$, using omega scans in the 20 range 5.2 to 50°. Three standard reflections measured after every 97 reflections collected exhibited less than 4% loss in intensity. The data were integrated and averaged to yield 1788 independent reflections ($R_{int} = 0.0233$).

The structure was solved by direct methods and refined by full-matrix least-squares techniques on F^2 using structure solution programs from the SHELXTL system.¹ The compound crystallized in the centrosymmetric triclinic space group P1bar (Z=2). The nonhydrogen atoms were refined anisotropically. Hydrogen atoms were located in the Fourier map and were refined isotropically. The final R factor values were $R_1=0.0308$ and $wR_2=0.0737$ for 1513 data with $F>4\sigma F$. ($R_1=0.0417$ and $wR_2=0.0792$ for all 1788 data giving a data to parameter ratio of 13:1 and a goodness of fit on F^2 of 1.069). The maximum and minimum residual densities remaining were 0.33 eÅ⁻³ and -0.30 eÅ⁻³, respectively. An absorption correction was not applied

Reference: Sheldrick, G. M.; SHELXTL Crystallographic System, Version 5.03/Iris (1995), Siemens Analytical X-ray Insts. Inc., Madison, WI.



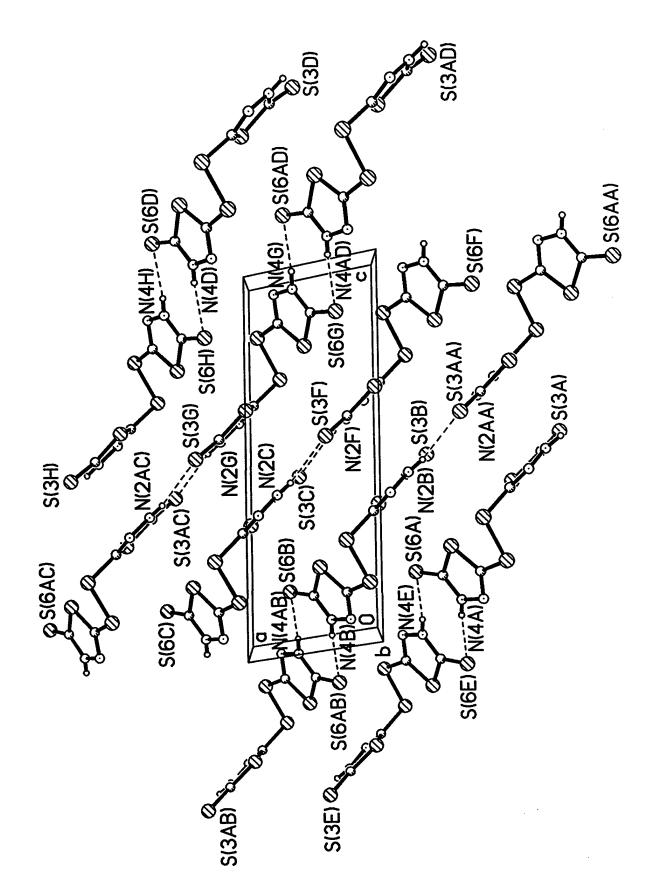


Table 1. Crystal data and structure refinement for 1.

Identification code	pag105b										
Empirical formula	C4H2N4S6										
Formula weight	298.46										
Temperature	293(2) K										
Wavelength	0.71073 Å										
Crystal system	Triclinic										
Space group	ΡĪ										
Unit cell dimensions	a = 5.7200(10) Å alpha = 94.410(10) b = 5.9170(10) Å beta = 98.440(10) c = 15.889(2) Å gamma = $104.460(10)$										
Volume, Z	511.46(14) Å ³ , 2										
Density (calculated)	1.938 Mg/m ³										
Absorption coefficient	1.297 mm ⁻¹										
F(000)	300										
Crystal size	0.27 x 0.22 x 0.06 mm										
θ range for data collection	2.61 to 25.00°										
Limiting indices	$-1 \le h \le 6$, $-6 \le k \le 6$, $-18 \le 1 \le 18$										
Reflections collected	2370										
Independent reflections	1788 (R _{int} = 0.0233)										
Refinement method	Full-matrix least-squares on F ²										
Data / restraints / parameters	1788 / 0 / 135										
Goodness-of-fit on F ²	1.069										
Final R indices $[I>2\sigma(I)]$	R1 = 0.0308, $wR2 = 0.0737$										
R indices (all data)	R1 = 0.0417, $wR2 = 0.0792$										
Largest diff. peak and hole	0.325 and -0.302 eÅ ⁻³										

Table 2. Atomic coordinates [\times 10⁴] and equivalent isotropic displacement parameters [$\mathring{\mathbf{A}}^2$ \times 10³] for 1. U(eq) is defined as one third of the trace of the orthogonalized V_{ij} tensor.

	x	У	Z	ប (eq)
S (1)	2500(1)	8480(1)	2697(1)	33 (1)
S(2)	-114(1)	4012(1)	3431(1)	35(1)
S(3)	-3950(2)	2145(1)	4511(1)	44 (1)
S(4)	762(1)	7046(1)	1474(1)	31(1)
S (5)	4509(1)	4253 (1)	2005(1)	34 (1)
S(6)	6931(1)	615 (1)	1365(1)	38(1)
N(1)	-836 (5)	8097 (4)	3715(2)	39(1)
N(2)	-2270 (5)	6576 (5)	4148(2)	40(1)
N(3)	2186(5)	4013 (4)	477 (2)	40(1)
N(4)	3652 (5)	2501(4)	486 (2)	37 (1)
C(1)	395 (5)	6988 (5)	3308(2)	31(1)
C(2)	-2246 (5)	4296 (5)	4078(2)	32(1)
C(3)	2450 (5)	5043 (4)	1243 (2)	27 (1)
C(4)	5028 (5)	2322(4)	1210(2)	28(1)

Table 3. Bond lengths [Å] and angles [O] for 1.

S(1)-C(1)	1.765(3)	S(1)-S(4)	2.0602(10)
S(2)-C(2)	1.737(3)	S(2)-C(1)	1.743(3)
S(3)-C(2)	1.660(3)	S(4)-C(3)	1.755(3)
S(5)-C(3)	1.736(3)	S(5)-C(4)	1.745(3)
S(6)-C(4)	1.666(3)	N(1)-C(1)	1.282(3)
N(1)-N(2)	1.360(4)	N(2)-C(2)	1.349(4)
N(2)-H(1)	0.84(4)	N(3)-C(3)	1.289(3)
N(3)-N(4)	1.370(3)	N(4)-C(4)	1.323(4)
N(4)-H(2)	0.79(3)		
C(1)-S(1)-S(4)	100.87(10)	C(2)-S(2)-C(1)	89.11(13)
C(3)-S(4)-S(1)	101.01(9)	C(3)-S(5)-C(4)	89.10(13)
C(1)-N(1)-N(2)	109.0(2)	C(2)-N(2)-N(1)	119.5(2)
C(2)-N(2)-H(1)	118(3)	N(1)-N(2)-H(1)	122(3)
C(3)-N(3)-N(4)	108.5(2)	C(4)-N(4)-N(3)	120.0(2)
C(4)-N(4)-H(2)	123 (3)	N(3)-N(4)-H(2)	117 (3)
N(1)-C(1)-S(2)	115.4(2)	N(1)-C(1)-S(1)	120.5(2)
S(2)-C(1)-S(1)	124.1(2)	N(2)-C(2)-S(3)	127.0(2)
N(2)-C(2)-S(2)	106.9(2)	S(3)-C(2)-S(2)	126.1(2)
N(3)-C(3)-S(5)	115.3(2)	N(3)-C(3)-S(4)	120.9(2)
S(5)-C(3)-S(4)	123.8(2)	N(4)-C(4)-S(6)	127.9(2)
D(3)=0(3)=D(4)	107.1(2)	S(6)-C(4)-S(5)	125.0(2)

Symmetry transformations used to generate equivalent atoms:

Table 4. Anisotropic displacement parameters $[\mathring{\mathbf{A}}^2 \times 10^3]$ for 1. The anisotropic displacement factor exponent takes the form: $-2\pi^2 \ [\ (\mathrm{ha}^*)^2 \mathbf{U}_{11} + \ldots + 2\mathrm{hka}^* \mathbf{b}^* \mathbf{U}_{12} \]$

	U11	U22	Ψ33	Ψ23	U13	U12
S (1)	38(1)	31(1)	31(1)	-1(1)	9 (1)	9(1)
S(2)	43 (1)	34(1)	32(1)	3(1)	15(1)	17 (1)
s (3)	53 (1)	42(1)	42(1)	4(1)	23 (1)	13 (1)
S(4)	32 (1)	35(1)	30(1)	2(1)	6 (1)	16(1)
s (5)	37(1)	39(1)	29(1)	-4(1)	1(1)	20(1)
S(6)	38(1)	38(1)	41(1)	-4(1)	2(1)	20(1)
N(1)	48 (2)	39(1)	38(1)	5(1)	15(1)	22(1)
N(2)	49 (2)	43 (2)	38(1)	4(1)	20(1)	22(1)
N(3)	50(2)	48 (2)	30(1)	1(1)	6(1)	30(1)
N(4)	48 (2)	41(1)	27 (1)	-2(1)	6(1)	24 (1)
C(1)	35 (2)	35(2)	26(1)	0(1)	6 (1)	14 (1)
C(2)	35 (2)	41(2)	22(1)	0(1)	5 (1)	13 (1)
C(3)	28(1)	26(1)	30(1)	4(1)	7 (1)	9 (1)
C(4)	27 (1)	25(1)	32(1)	-1(1)	8(1)	4(1)

Table 5. Hydrogen coordinates (\times 10⁴) and isotropic displacement parameters ($\mathring{A}^2 \times 10^3$) for 1.

	x	У	Z	Ŭ(eq)		
H(1)	-3310(71)	6967 (68)	4409 (25)	72 (13)		
H(2)	3631(63)	1777 (58)	47 (22)	50(10)		

49257441336253144314152435374312634164234143544272322414344352223355533323512136470244263

10s

64441365724521536194474335663486556322333543484212023355572224482542244352653432767393542

01234565432101234565432101234566543210123456654321123456654321123456654321012345665432101234566543210

10Fo 10Fo 10Fc 10s 10s 10Fo 10Fc 10s 10Fc 10s h k l 10Fo 10Fc 10s l 10Fo 10Fc h 549554642584614125444453657310443443156023571544747555135474245963674433554753784336443773 12345654527012346545270123545270124527012457701234527701234545452701234565452701245654527012345 26480647664394343552332731453742563555556433355442824317550422523364694344274631756325444 665432101234566543210123456654321012345654321012345654321012345654321012346545432101234654321012354321012354321012 75979565876754515826210779466523302413122790246262625513097731367880589458754589199604475848116774733 123437101234543710123455437101234565437101234565437101234566543710123456543710123456543710123456543710123456543 3840 19267459990569550255250138400511298803271325818812873552884628339558866496333195691346637456839558617818461 1853342885300027335011931389777552566091257346109666018812274417415772953746860257451911489314893388775525860912338877336018812274414129557368802573594519921348923888741178876675434 195638026282882026551036604090074411102228759433917522306097518829257437928684401288655193424288727209 195638026288202655103660409900744111022287594339175223060975189129257435792858240121833727209 324633464137624343464864532341334777781442134227812435179342253447352344543312374321649 5665432101234565432101234565432101234565432101234654321012346543210123465432101235432101243210123 46316431745512953543210353345452532271034365228333444336223333944474539349134239455303532 531351314363501336352254382336533462387542544835464134442451154373393830535975447892237

l 10Fo 10Fc 10s l 10Fo 10Fc 10s k l 10Fo 10Fc 10s h k 10Fo 10Fc 10s h h 3432331411361231236354044536423895423575131454384235833426782624310442536158442354343533 01234554321012345543210123456543210123456543210123465432101234654321012346543210123465432101235432101235 938992883611850144505447261882988771884818724444441141555533099115005216664004499781683399135836118501445005449983913991384187244444411155553309911500521666400449978168399913583 5432101234565432101234565432101234654321012354321012543210132110123210123443210123454321 655314433532345411556453136934976312531141542222253146341<mark>0</mark>633441423321214014256360353535473 3823571430416255561145421733415363632541135742341747339252516153354243775368493442221433 5432101234654321012354321013211012332101234432101234554321012345543210123456543210123456 34352333146313465145774332033304319142448993315223314043425452437163226473481343014333344 3210123443210123455432101234554321012345555321012345654321012345654321012345654321012345654321012345654321012345

h l 10Fo 10Fc 10s 53363300280644470804586577794024455545531633998195545252588882444355188725789135944355443455545278828887158887257898145559445554553199811555452578824887257888715888725789871567399313251119655 65518019136264631574235321635745133633813327524394524446386354744124365855416253334442256

10s

42375343447590155742354624520437521110336624457365642866453262437443255321153535252234632

l 10Fo 10Fc 10s l 10Fo 10Fc 10s l 10Fo 10Fc 10s h k h l 10Fo 10Fc ħ k h k l 10Fo 10Fc 10s h k k 24810242303758995111783995111844824442558501766666977221175125462014547733088555097004946004440053345567 28104994240480779290480079595556655793734115175773223334873791192771593856102715929171172345783912407155291711923447715529171172534548 82414977587320548813555212688335505062447 29099697132772688587311531657368833024447920118713183120 3 2 21311 10 5 33755062447 29099697132773587 113457782311665786833024447921037735587 21001210123210123432101234321012354321012354321012354321012354321012432101243210132121012343210123 54321012543210132121012332101234321012344321012345432101234543210123454321012345432101234543210123454321012345 517133324184342341394134585523435309535543232323464372743334543533344442582333173887372345 473272075624624264943248114435215433432474495308444438346416430553343633137422233333520 55525421855338222548303534745392413413425553343363314634642813329544563345153304454365877 452101234321012343210124321014521010121012321012432101243210124321012432101243210143210132110121013210 849444574113725585334649141357433324463434123454305412802533131223535353535354135633135733 163 464 1636 1167 721 183 97 28 150 157 75 1145 227 93 57 231 96

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